

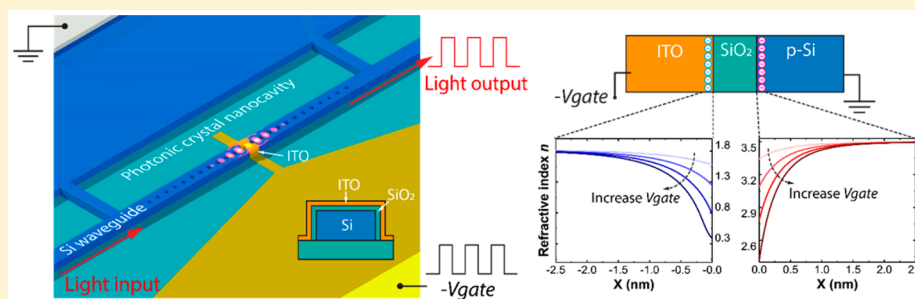
1 Ultracompact Silicon-Conductive Oxide Nanocavity Modulator with 2 0.02 Lambda-Cubic Active Volume

3 Erwen Li,[†] Qian Gao,[†] Ray T. Chen,[‡] and Alan X. Wang^{*,†}

4 [†]School of Electrical Engineering and Computer Science, Oregon State University, Corvallis, Oregon 97331, United States

5 [‡]Department of Electrical and Computer Engineering, The University of Texas at Austin, Austin, Texas 78758, United States

6 **S** Supporting Information



7 **ABSTRACT:** Silicon photonic modulators rely on the plasma dispersion effect by free-carrier injection or depletion, which can
8 only induce moderate refractive index perturbation. Therefore, the size and energy efficiency of silicon photonic modulators are
9 ultimately limited as they are also subject to the diffraction limit. Here we report an ultracompact electro-optic modulator with
10 total device footprint of $0.6 \times 8 \mu\text{m}^2$ by integrating voltage-switched transparent conductive oxide with one-dimensional silicon
11 photonic crystal nanocavity. The active modulation volume is only $0.06 \mu\text{m}^3$, which is less than 2% of the lambda-cubic volume.
12 The device operates in the dual mode of cavity resonance and optical absorption by exploiting the refractive index modulation
13 from both the conductive oxide and the silicon waveguide induced by the applied gate voltage. Such a metal-free, hybrid silicon-
14 conductive oxide nanocavity modulator also demonstrates only 0.5 dB extra optical loss, moderate Q-factor above 1000, and high
15 energy efficiency of 46 fJ/bit. The combined results achieved through the holistic design opened a new route for the development
16 of next generation electro-optic modulators that can be used for future on-chip optical interconnects.

17 **KEYWORDS:** Silicon photonics, transparent conductive oxides, optical modulator, photonic crystal cavity, plasmonics

18 **T**he ever-increasing demand to process, store, and exchange
19 information creates an unceasing driving force for high-
20 bandwidth, energy-efficient photonic technologies. In recent
21 years, the vision to develop photonic devices with extremely
22 high energy efficiency to attojoule/bit has been outlined.^{1,2}
23 Silicon photonics has the potential to transform future optical
24 interconnect systems by reducing the energy consumption and
25 enhancing the bandwidth of existing electronic systems by
26 orders of magnitude using complementary metal-oxide-semi-
27 conductor (CMOS) compatible fabrication processes.^{3–5} For
28 example, silicon electro-optic (E-O) modulators have been
29 reported with femtojoule/bit energy efficiency.^{6,7} In addition to
30 the application in optical interconnects, silicon photonic
31 devices can also operate the logic gates to conduct certain
32 types of optical computation.^{8–10} However, the performance of
33 silicon photonic devices is still limited by the diffraction limit
34 and the relatively weak plasma dispersion effect. Although
35 silicon has a relatively high refractive index, it can only shrink
36 the wavelength inside the silicon waveguide proportionally to
37 the scale of λ/n , roughly to 400–600 nm. Further reduction of
38 the device footprint requires exploiting surface plasmon
39 polaritons (SPPs), which are bound waves at the interface

between a metal and a dielectric.¹¹ The extremely strong light
40 confinement of metal–insulator–metal (MIM) waveguide has
41 led to the demonstration of ultracompact and high-bandwidth
42 plasmonic E-O modulators.^{12,13} However, plasmonic structures
43 and devices are very lossy and can only carry information over a
44 very short distance. Therefore, hybrid plasmonic-dielectric
45 waveguide integration must be used for real optical
46 interconnects,¹² which increases the complexity of design and
47 fabrication.

The second constraint of silicon photonic devices is the
48 plasma dispersion effect induced by free-carrier injection or
49 depletion,¹² which can only induce moderate refractive index
50 perturbation. For example, for a typical depletion-based silicon
51 photonic modulator with a moderate doping level of 2.5×10^{18}
52 cm^{-3} in its active region,⁶ when it is completely depleted, the
53 refractive index only changes by 0.06%. As a result, current
54 Mach–Zehnder interferometer (MZI) silicon modulators
55 require a long device length up to hundreds of micrometers
56
57

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58 to several millimeters to accumulate sufficient phase modu-
 59 lation.¹⁴ The large device footprint also leads to a large energy
 60 consumption of picojoule/bit, which cannot meet the require-
 61 ment of future photonic interconnects application. Compared
 62 with MZI modulators, resonator-based E-O modulators occupy
 63 a much smaller footprint and achieve significantly higher energy
 64 efficiency. To date, various ultracompact silicon microring
 65 resonators,^{15–17} microdisks resonators,^{6,18} and photonic crystal
 66 nanocavity¹⁹ have been demonstrated and used in optical
 67 interconnect systems, achieving high performance in modu-
 68 lation speed, compactness, and energy efficiency. However,
 69 resonator-based modulators have an intrinsic trade-off between
 70 energy efficiency and optical bandwidth. For practical devices,
 71 thermal control with integrated heater and temperature sensors
 72 are often used to obtain stable performance,^{20,21} but with the
 73 sacrifice of additional energy consumption and footprint.
 74 To overcome the intrinsic drawback of the plasma dispersion
 75 effect of silicon, various functional materials, such as
 76 graphene,^{22,23} vanadium oxide,²⁴ and ferroelectric materials²⁵
 77 have been integrated with silicon photonics to build next
 78 generation E-O modulators. Among all these emerging
 79 materials, transparent conductive oxides (TCOs) have attracted
 80 escalating interests as a new type of plasmonic material^{26,27} and
 81 as active materials for E-O modulators^{28–31} in recent years due
 82 to the large tunability of their refractive indices. TCOs, such as
 83 indium–tin oxide (ITO) and aluminum–zinc oxide (AZO), are
 84 a family of wide-bandgap semiconductor oxide materials that
 85 can be degenerately doped to a high level, which is widely used
 86 in the display industry.³² With free-carrier concentrations
 87 ranging from 1×10^{19} to 1×10^{21} cm⁻³, the real part n of the
 88 refractive index could experience more than 1 refractive index
 89 unit (RIU) change,³³ as shown in Figure 1a. Meanwhile, the
 90 imaginary part κ increases to the same order of magnitude as
 91 the real part, which causes dramatic increase of the absorption
 92 30–140× larger than that of silicon, as shown in Figure 1b. In
 93 recent years, a unique property called epsilon-near-zero (ENZ)
 94 is verified with TCO materials.^{34,35} At very high free-carrier

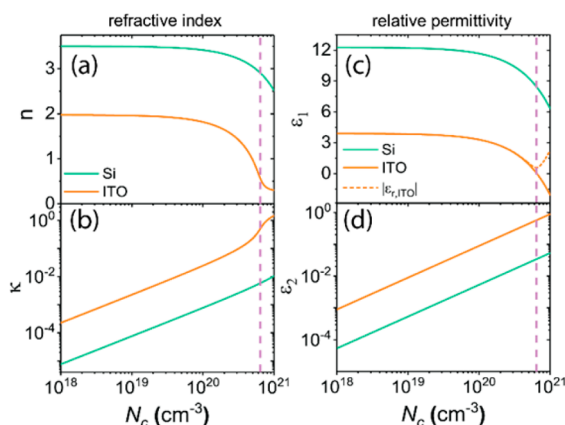


Figure 1. (a) Real part (n) and (b) imaginary part (κ) of the refractive indices of p-type Si (green solid) and ITO (orange solid) as a function of free-carrier concentration N_c (hole in Si, $N_{h,Si}$; and electron in ITO, $N_{e,ITO}$) at wavelength $\lambda = 1.55$ μm . (See the Supporting Information for calculation details.) (c) Real part (ϵ_1) and (d) imaginary part (ϵ_2) of the relative permittivity of p-type Si (green solid line) and ITO (orange solid line) as a function of N_c at wavelength $\lambda = 1.55$ μm . The orange dashed line in part c shows the absolute permittivity of ITO ($|\epsilon_{r,ITO}|$), and the pink dashed line indicates the N_c where the ITO reaches ENZ.

concentration, the real permittivity of TCOs reaches zero while
 the absolute permittivity is a minimum value due to the small
 value of the imaginary part as indicated by the vertical dotted
 lines in Figure 1c,d. In this case, the electric field will be
 strongly confined in TCOs due to the continuity of electric
 field displacement at the material interface. ENZ will further
 enhance the light–matter interaction as discussed in ref 36. For
 silicon, however, it is still far from ENZ even at 10^{21} cm⁻³ free-
 carrier concentration due to the large value of its high
 frequency permittivity.

Existing TCO-based E-O modulators are exclusively based
 on straight silicon waveguide^{28–30} or plasmonic slot wave-
 guide³¹ using electrically induced optical absorption from the
 integrated MOS capacitor. The phase change induced by the
 real part of the permittivity of the TCO materials, although
 automatically accompanying the imaginary part of the index
 change, does not contribute to any E-O modulation.
 Therefore, a relatively long modulation length (a few microns)
 is required to induce sufficient optical absorption. Moreover,
 these TCO modulators require the presence of metal gates for
 strong plasmonic light confinement and electronic signal
 conductance, which introduce relatively high optical loss even
 at the transparent state. In this manuscript, we present an
 ultracompact hybrid silicon-TCO nanocavity modulator to
 overcome the intrinsic drawbacks of those straight waveguide
 modulators. There are two exclusive advantages compared with
 existing TCO-based modulators. First, the active region of our
 plasmonic E-O modulator is free of metal. The metal gate of
 the MOS capacitor is replaced by an ITO gate, which induces
 much smaller optical absorption compared with other metal-
 gated modulators. This ITO–oxide–Si capacitor offers the
 possibility to build a relatively high Q -factor resonator while
 traditional metal–oxide–ITO cannot. Second, in our nano-
 cavity E-O modulator, both the phase change and the
 absorption, from both the Si and ITO materials, will contribute
 coherently to E-O modulation. The total device footprint of our
 TCO modulator is only 0.6×8 μm^2 using one-dimensional
 (1D) photonic crystal (PC) nanocavity with 20 nm SiO₂ as the
 insulator and 20 nm ITO as the gate. The E-O modulation
 volume is less than 0.06 μm^3 (width \times height \times length = 0.56
 $\mu\text{m} \times 0.28$ $\mu\text{m} \times 0.375$ μm), namely, only 2% of lambda-cubic
 (0.02 λ^3) volume, which is the smallest active modulation region
 that has ever been reported to the best of our knowledge. The
 E-O modulation volume is the most critical device metric that
 affects the energy efficiency of an E-O modulator,¹ which is
 usually achieved by compact resonant cavities or plasmonic
 structures. A few ultracompact resonator-based E-O modulators
 have been reported, including microdisk modulators^{6,18} using
 vertical p–n junction with an active volume of 1.6–2.5 μm^3 and
 p–i–n photonic crystal nanocavity modulator¹⁹ with a
 modulation volume of 2.2 μm^3 . Non-resonator-type TCO
 plasmonic modulators have typical lengths of 5 μm ³⁰ to 10
 μm ³¹ long, with calculated active modulation volume around
 0.6 μm^3 . Our device combines the advantages of ultracompact
 resonators and TCO plasmonics, which further reduces the
 active E-O modulation volume by 10×.

Briefly, the applied gate voltage induces free electron and
 hole accumulation ITO and silicon, respectively. The free-
 carrier-induced variation of the real part of the optical
 permittivity causes blue-shift of the resonance peak, while the
 increase of the imaginary part of the optical permittivity induces
 optical absorption of the resonance mode, which becomes
 more prominent when ITO is close to ENZ. We experimentally

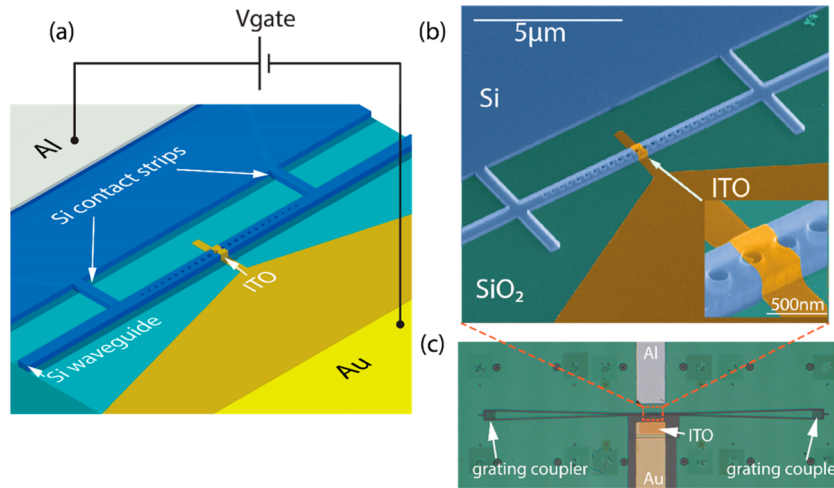


Figure 2. (a) 3D schematic of the Si-ITO modulator. (b) Colored scanning electron micrograph (SEM) of the fabricated Si-ITO modulator. The insertion figure shows the zoomed-in view of the center of the MOS capacitor region. (c) Optical image of the fabricated modulator.

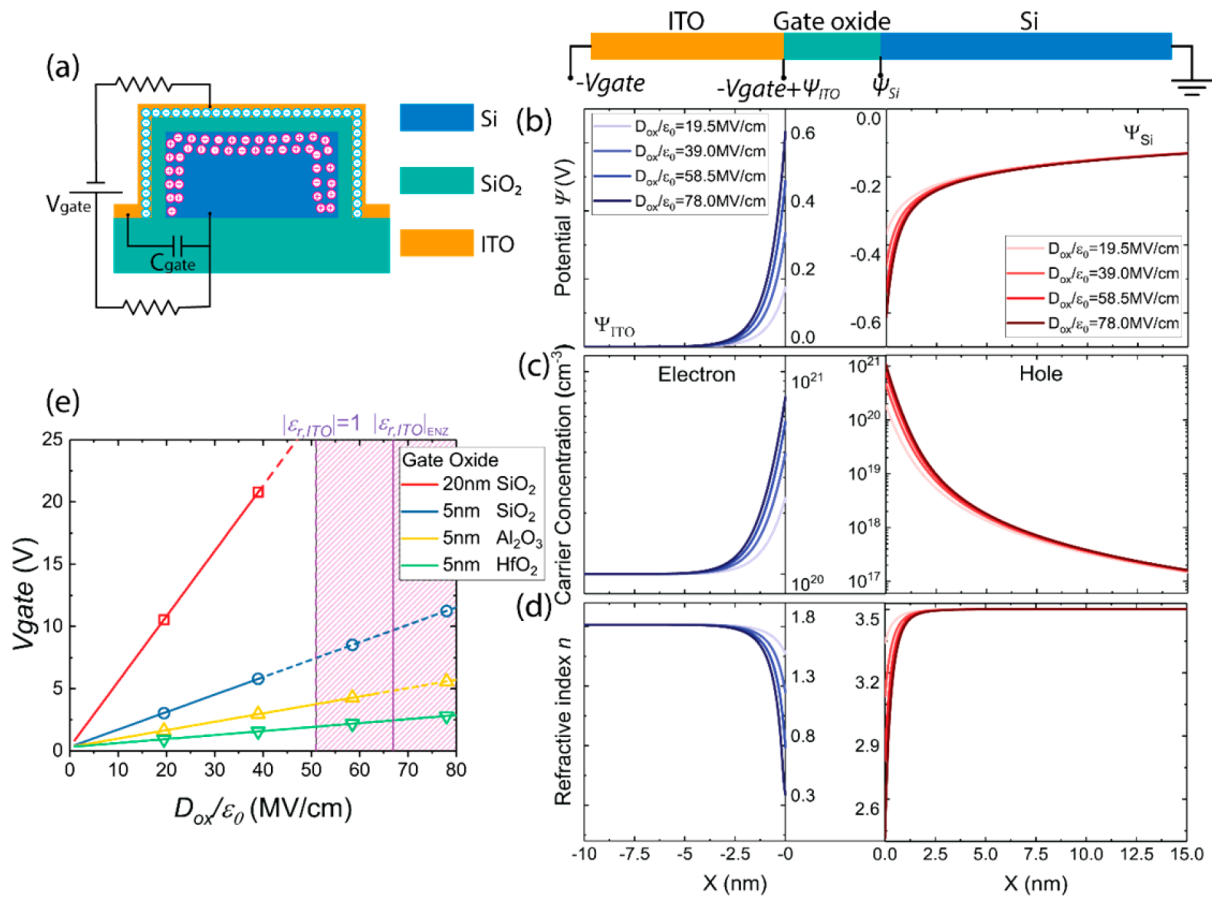


Figure 3. (a) Cross-section of the Si/oxide/ITO MOS capacitor at the center of the hybrid Si-ITO modulator. When a negative bias is applied on the ITO gate, electrons and holes accumulate at the ITO/oxide and Si/oxide interfaces, respectively. (b) Electrical potential distribution in ITO (blue lines) and Si (red lines) as a function of electrical displacement field in the gate oxide layer, D_{ox} . (c) Carrier density distribution in ITO (electron) and Si (hole) as a function of D_{ox} field. (d) The real part refractive index (n) distribution in ITO and Si as a function of D_{ox} field. (e) Gate voltage as a function of the D_{ox} field for different gate oxide layers: 20 nm SiO_2 (red line), 5 nm SiO_2 (blue line), 5 nm Al_2O_3 (yellow line), and 5 nm HfO_2 (green line). The dashed lines show the D_{ox} field range when the gate oxide layer will breakdown. The shaded area enclosed by the purple dashed line shows the D_{ox} field range when the permittivity of ITO accumulation layer, $|\epsilon_{r,ITO}|$, is smaller than 1, representing the ENZ region; the purple solid line indicates the D_{ox} field when $|\epsilon_{r,ITO}|$ reaches minimum ENZ value.

158 achieved a large E-O response of 30 pm/V and high energy
 159 efficiency of 46 fJ/bit. Compared with those of reported TCO-
 160 based plasmonic modulators, the active region of our device is

completely free of metallic materials, which offers a low device 161
 loss of only 0.5 dB, moderately high Q-factor of 1000, and 162
 better compatibility with CMOS processes. Compared with the 163

164 conventional silicon ring resonator or microdisk modulator, our
 165 device shows exclusive advantages as it provides a larger
 166 resonant wavelength tuning and much higher usable optical
 167 bandwidth of greater than 1 nm. Through future research by
 168 replacing the current SiO₂ gate with high-*k* materials and
 169 improving the *Q*-factor, we can potentially achieve even higher
 170 energy efficiency below 1 fJ/bit.

171 The schematic of the ITO-gated 1D silicon PC nanocavity is
 172 shown in Figure 2a. The device consists of a MOS capacitor
 173 built at the center of the nanocavity on a silicon strip
 174 waveguide. The strip waveguide is fabricated on a p-type
 175 silicon-on-insulator (SOI) substrate with 500 nm in width and
 176 250 nm in height. A pair of grating couplers are integrated to
 177 couple light in and out of an optical fiber. The PC cavity is
 178 defined through electron beam lithography (EBL) and reactive
 179 ion etching (RIE), operating in the TE mode. Two photonic
 180 crystal mirror segments are placed back-to-back adjacent to the
 181 nanocavity. The air hole size is quadratically tapered down from
 182 the center of the cavity region to the edge of the two mirror
 183 segments. In our design, each mirror segment has 12 air holes.
 184 The filling factor, which is defined as $f = A/pw$, is tapered down
 185 from 0.23 in the center to 0.1 at the edge, where *A* is the air
 186 hole area, *p* is the air hole period, and *w* is the waveguide width.
 187 The period *p* is chosen to be 340 nm to allow the modulator to
 188 operate in the telecommunications wavelength range. In the
 189 center of the cavity, an ITO/SiO₂/Si film stack creates a MOS
 190 capacitor with cross-sectional view shown in Figure 3a. Here,
 191 the silicon waveguide also serves as the bottom electrode
 192 despite its relatively high resistivity. Two 400 nm wide silicon
 193 strips are used to form the conduction path between the silicon
 194 waveguide and the silicon slab with the contact electrodes.
 195 Then, a 20 nm thick SiO₂ layer is thermally grown on top of the
 196 entire silicon PC nanocavity serving as the gate oxide. Finally, a
 197 20 nm thick ITO layer is sputtered, performing as the metallic
 198 gate electrode. We need to emphasize that the center
 199 nanocavity length is only 120 nm, which is at least 50× shorter
 200 than ring resonators or microdisk resonators. A 375 nm long
 201 ITO gate is made to compensate the misalignment of the
 202 electron beam lithography (EBL) process as shown by the inset
 203 figure of Figure 2b. The SEM and optical images of one
 204 fabricated device are depicted in Figure 2b,c (see the
 205 Supporting Information for details of fabrication).

206 The device operates in the accumulation mode of the MOS
 207 capacitor with the negative gate bias on the ITO gate. Unlike
 208 other reported TCO-MOS E-O modulators which ignore the
 209 free-carrier effect in the metal gate, we consider the free-carrier
 210 accumulation at both sides of the interfaces, i.e., in the ITO/
 211 SiO₂ and Si/SiO₂ interfaces. We perform a numerical
 212 simulation systematically to analyze the carrier distribution in
 213 the accumulation layers versus the applied gate bias. In our
 214 modeling, the carrier density and electric potential in the ITO
 215 and Si regions are treated in different ways. The main difference
 216 is that the high doping level of ITO results in an initial Fermi
 217 level higher than the bottom of the conduction band.
 218 Therefore, the electron density and electric potential in ITO
 219 behave more like a metal, which can be approximated by the
 220 Thomas–Fermi screening model.^{37,38} On the other side, Si
 221 follows the classic semiconductor theory.³⁹ However, a large
 222 band bending is expected in our device, and a traditional
 223 Boltzmann distribution approximation is not accurate. A
 224 rigorous analysis using the Fermi–Dirac distribution is used
 225 to model the Si side. In order to obtain representative results,
 226 we conduct our modeling using the electric displacement field

*D*_{ox} instead of the electric field *E*. The boundary condition only
 227 requires the value of *D*_{ox} in the gate oxide layer, making the
 228 modeling independent of the gate oxide material and thickness.
 229 We plot the electric potential and carrier distribution as a
 230 function of *D*_{ox} as shown in Figure 3b,c. We can see that the
 231 electron concentration in ITO (*N*_{e,ITO}) accumulates from 1 ×
 232 10²⁰ to 7.46 × 10²⁰ cm³, and the hole concentration in Si (*N*_{h,Si})
 233 accumulates from 1 × 10¹⁷ to 1.08 × 10²¹ cm³ with a *D*_{ox}/*ε*₀
 234 value of 78 MV/cm. Surprisingly, the peak of *N*_{h,Si} is even
 235 higher than that of *N*_{e,ITO}, which is because of the larger
 236 effective density of state of Si compared with that of ITO (see
 237 the Supporting Information). As a result, *N*_{h,Si} in Si is more
 238 sensitive to electrical potential modulation than *N*_{e,ITO} in ITO.
 239 The ITO reaches the ENZ region when the *N*_{e,ITO} is 6.4 × 10²⁰
 240 cm³ with *D*_{ox}/*ε*₀ of 67 MV/cm. Figure 3d plots the
 241 corresponding distribution of the refractive indices of ITO
 242 and Si. Both ITO and Si exhibit dramatic refractive index
 243 modulation within a thin layer of ~1 nm thick close to the
 244 interface even at a relatively small *D*_{ox} field. For the ITO side,
 245 the effect of this thin accumulation layer is already well-
 246 recognized.^{30,37,38} This layer is often treated as an effective
 247 accumulation layer, and the thickness can be estimated by the
 248 Thomas–Fermi screening length, *L*_{tf}. On the Si side, this thin
 249 accumulation layer could also play a critical role for the E-O
 250 modulation but was not utilized by simple straight waveguides
 251 in published papers. Detailed analysis will be provided in the
 252 following section. Next, knowing the *D*_{ox} field, we can calculate
 253 the gate voltage by $V_{\text{gate}} = |\Psi_{\text{ITO}}| + \frac{D_{\text{ox}} t_{\text{ox}}}{\epsilon_0 \epsilon_{\text{oxide, st}}} + |\Psi_{\text{Si}}|$, where Ψ_{ITO}

254 and Ψ_{Si} are the surface potential at the ITO/SiO₂ and the Si/
 255 SiO₂ interface, *ε*₀ is the vacuum permittivity, and *ε*_{oxide, st} and *t*_{ox}
 256 are the static relative permittivity and thickness of the gate
 257 oxide layer. Figure 3e plots the applied gate voltage as a
 258 function of *D*_{ox} field with different oxide materials and
 259 thicknesses. Here the dashed lines indicate a large *D*_{ox} field
 260 exceeding the breakdown of the gate oxide. From this analysis,
 261 it is obvious to draw a conclusion that thinner oxide layer
 262 thickness and high-*k* materials will help to reduce the applied
 263 bias voltage. Besides, to truly reach the ENZ operation of the
 264 ITO layer, a high-*k* gate material such as HfO₂ is necessary. In
 265 our experimental demonstration, we chose SiO₂ as the gate
 266 oxide material primarily due to our current fabrication facilities.
 267

268 The Si-ITO nanocavity modulator operates in the dual mode
 269 of cavity resonance and optical absorption. At a relatively small
 270 applied bias, the device operates in the “normal mode”, when
 271 the *N*_{e,ITO} is not high enough to push ITO into the ENZ
 272 confinement. Modulation of the nanocavity resonance domi-
 273 nates, which mainly comes from the real parts of the
 274 permittivity change ($\Delta\epsilon_1$) induced by the plasma dispersion
 275 effect of the ITO and Si. Based on the cavity perturbation
 276 theory, the resonance shift ($\Delta\omega$) can be expressed as⁴⁰

$$\Delta\omega = \frac{-\frac{\omega}{2} \int \Delta\epsilon E^* E \, dv}{\int \epsilon E^* E \, dv}$$

277 where ω is the original resonance frequency, ϵ and $\Delta\epsilon$ are the
 278 distribution of the original and changed permittivity, and *E* is
 279 the electric field distribution of the cavity mode. We know that
 280 the permittivity change caused by the plasma dispersion is
 281 proportional to the change of free-carrier concentration,
 282 namely, $\Delta\epsilon \propto \Delta N_c$. This means that the resonance shift
 283 induced by a 1 nm thick accumulation layer with a *N*_c of 1 ×
 284 10²⁰ cm⁻³ is equivalent to the shift induced by a 100 nm thick

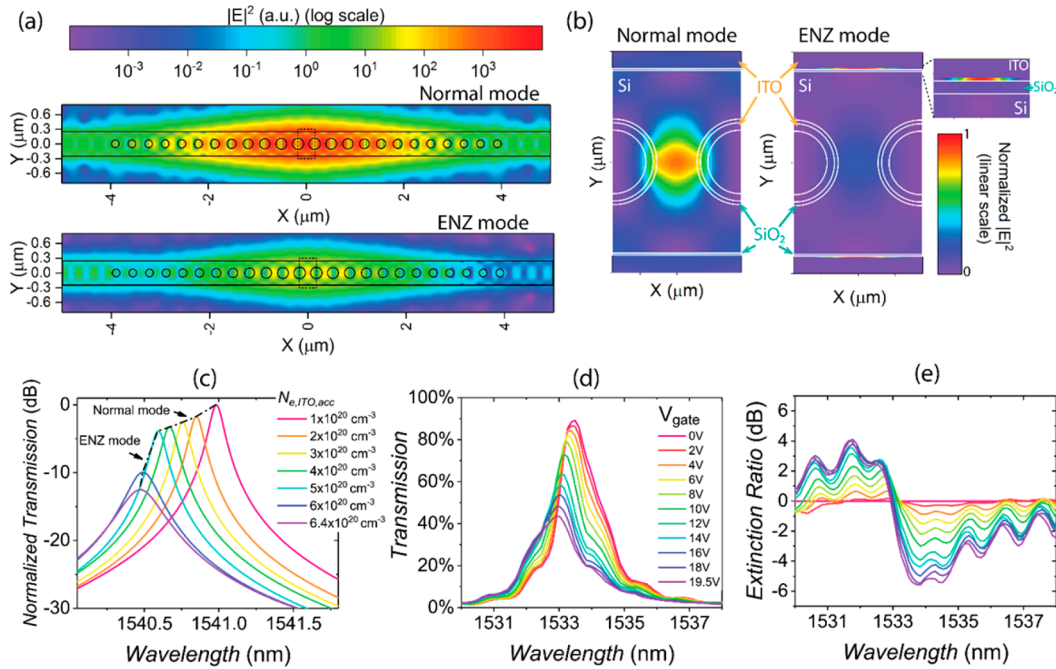


Figure 4. (a) Photonic crystal cavity mode profiles of “normal mode” (accumulation layer $N_{e,ITO} = 1 \times 10^{20} \text{ cm}^{-3}$) and “ENZ mode” (accumulation layer $N_{e,ITO} = 6.4 \times 10^{20} \text{ cm}^{-3}$). The optical field intensity is plotted in log scale. Clearly, at “ENZ mode” the transmission drops due to the ITO absorption. (b) Zoomed-in mode profile of “normal mode” and “ENZ mode”. The optical intensity is plotted in normalized linear scale. Inset: further zoomed-in mode profile of “ENZ mode” at the ITO/SiO₂ interface. It is clearly shown that in “ENZ mode” the optical field is strongly confined in the accumulation layer at the side wall. (c) Simulated normalized transmission spectrum at different free-carrier concentration $N_{e,ITO,acc}$ in the ITO accumulation region. The black dashed line outlines the change of the transmission peak as $N_{e,ITO}$ increases. (d) Measured static transmission spectrum as a function of the applied bias voltage. The DC applied bias ranges from 0 to 19.5 V. (e) Measured extinction ratio (ER) spectrum as a function of the applied bias voltage.

285 layer from full depletion to a N_c of $1 \times 10^{18} \text{ cm}^{-3}$ under the
 286 uniform optical field distribution approximation. Figure 4a,b
 287 shows the simulated photonic crystal cavity mode profile. The
 288 cavity mode has a good overlap with the accumulation layer of
 289 the MOS structure near the center air holes and is relatively
 290 uniform. Thus, it is reasonable to assume an approximately
 291 uniform optical distribution here. The resonance shift has the
 292 following relationship:

$$\Delta\omega \propto \frac{\omega \int \Delta N_c dv}{\epsilon_{eff} v_c} = \frac{\omega \Delta Q}{\epsilon_{eff} v_c} = \frac{\omega CV}{\epsilon_{eff} v_c} = \frac{\omega CV}{\epsilon_{eff} \gamma v_a} \propto \frac{C}{v_a}$$

293 where ϵ_{eff} and v_c are the effective permittivity and mode volume
 294 of the cavity mode, ΔQ is the accumulated free carriers induced
 295 by the applied voltage V , C and v_a are the capacitance and
 296 volume of the active modulation region of the modulator,
 297 respectively, and γ is the coefficient describing the overlapping
 298 between v_a and v_c . Additionally, due to the small mode volume
 299 of the photonic crystal cavity mode and its large overlap with
 300 the active modulation region of the modulator (Figure 4b), we
 301 can conclude that the resonance shift is proportional to the
 302 capacitance per unit active volume. Large capacitance C and
 303 small active volume v_a are preferred for high modulation
 304 efficiency. Since we effectively construct a 3D MOS capacitor in
 305 the center of the photonic crystal cavity, free carriers
 306 accumulate at all three interfaces. As large C/v_a ratio is realized,
 307 we can achieve significant resonance modulation within $0.02\lambda^3$
 308 active modulation volume. In spite of the resonance shift
 309 induced by the real part permittivity change, the optical
 310 absorption from the imaginary part change of the permittivity,
 311 which is usually a minor effect in pure silicon modulators, also

plays an important role in the Si-ITO hybrid modulator 312
 because of the 30–140× larger imaginary part of ITO 313
 compared with Si. As a result, larger extinction ratio can 314
 be achieved at the same resonance tuning. As the applied bias 315
 increases, the accumulation layer of ITO approaches the ENZ 316
 region as shown by the shaded area in Figure 3e. Once the 317
 modulator reaches the “ENZ mode”, the optical mode starts to 318
 be confined in the ITO accumulation layer. This ENZ 319
 confinement effect is highly polarization sensitive. For our 320
 photonic crystal nanocavity design operating in the TE mode, it 321
 mainly happens at the sidewall interface as shown in Figure 4b. 322
 The ENZ confinement effect will dramatically enhance the 323
 absorption which is proportional to $\frac{\epsilon_{2,ITO}}{2|\epsilon_{ITO}|^2}$.⁴¹ In this case, the 324

optical absorption mode dominates. Figure 4c plots the 325
 simulated transmission spectra of the hybrid Si-ITO modulator 326
 at different carrier concentrations in the accumulation region, 327
 $N_{e,ITO,acc}$. The black dashed line outlines the evolution of the 328
 transmission peak. The trend from the normal resonance 329
 modulation to ENZ electroabsorption is clearly shown as $N_{e,ITO}$ 330
 increases. 331

The E-O modulation response of fabricated hybrid Si-ITO 332
 modulator was characterized (see the Supporting Information 333
 for details of measurement setup). Figure 4d shows the 334
 measured transmission spectra as a function of the applied bias. 335
 The spectra are normalized to a straight Si waveguide as the 336
 reference. The insertion loss (IL) of the PC nanocavity 337
 modulator is only 0.5 dB at the peak resonance wavelength. 338
 The free-carrier concentration of as-sputtered ITO is 1×10^{20} 339
 cm^{-3} , which is still a dielectric material at telecommunications 340
 wavelengths. The measured Q -factor after ITO deposition is 341

342 around 1000, which is slightly smaller than the Q -factor
 343 measured before sputtering the ITO (~ 1200), proving that the
 344 degradation of the Q -factor due to the thin ITO layer is minor.
 345 The resonance wavelength blue-shifts by 0.57 nm with a change
 346 in DC bias from 0 to -19.5 V, indicating a 30 pm/V
 347 modulation efficiency. Meanwhile, we observe a significant drop
 348 of the peak transmission by 45.34%, which is caused by the
 349 resonance shift as well as the optical absorption. The MOS
 350 capacitor operation is verified by the low leakage current, which
 351 is measured to be less than 100 fA at -20 V. Figure 4e plots the
 352 extinction ratio (ER) spectrum as a function of the applied bias.
 353 A usable optical bandwidth of greater than 1 nm is observed if
 354 we allow 1 dB variation of the ER. The maximum modulation is
 355 observed at 1533.78 nm, which introduces an additional loss of
 356 0.75 dB as compared to the peak wavelength. The transmission
 357 varies by 5.6 dB with a bias changing from 0 to -19.5 V. The
 358 dynamic modulation speed is demonstrated up to 3.2 MHz
 359 with an AC voltage swing of 0 to -12 V (as shown in Figure 5),
 360 which is limited by our testing instruments.

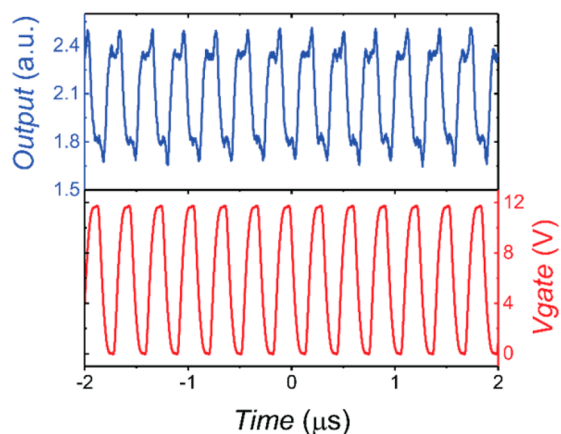


Figure 5. AC optical modulation testing results at 1534.78 nm with 0 to -12 V sweep input bias voltage at 3.2 MHz.

361 Here we estimated the modulation speed and energy
 362 efficiency of the hybrid Si-ITO nanocavity modulator. The
 363 speed of the modulator is limited by the RC delay since its
 364 operation is based on the fast accumulation mode of a MOS
 365 capacitor. The finite element method (FEM) simulation gives
 366 the capacitance of the modulator including the whole PC
 367 nanocavity and the ITO gate in the active region to be 1.28 fF.
 368 The series resistance of our fabricated device is around 4.9 M Ω ,
 369 which is limited by the lightly doped (1×10^{15} cm $^{-3}$) SOI slab.
 370 Consequently, our current device has a relatively slow RC-
 371 limited speed of 160 MHz. However, the series resistance can
 372 be reduced to ~ 9 K Ω by selectively doping the silicon
 373 conduction strips and PC waveguide to a high level of 5×10^{18}
 374 cm $^{-3}$ while keeping the doping of the center active cavity
 375 region at a moderate high level of 1×10^{17} cm $^{-3}$ (see the
 376 Supporting Information for details of capacitance and resistance
 377 calculation). The optical loss of a passive silicon waveguide with
 378 high-level doping is around 0.017 dB/ μ m according to our
 379 optical FEM simulation. A 10 μ m long silicon waveguide with
 380 high doping level will only introduce an additional loss of 0.17
 381 dB. Besides, the corresponding silicon waveguide loss of
 382 moderate high doping level is 3.4×10^{-4} dB/ μ m. For a cavity
 383 with a moderate Q -factor of 5000, which corresponds to a
 384 photon lifetime of 4.2 ps, the increasing in optical loss is only

0.12 dB. As a result, the RC-limited bandwidth can be
 improved to 87 GHz. However, the real achievable operation
 speed will be limited by the electronic circuit or signal
 generator. The energy efficiency of the modulator is estimated
 using $E_{\text{per-bit}} = CV^2/4$. Assuming a 12 V voltage swing (3 dB ER
 at the resonance peak), the energy consumption of the device is
 only 46 fJ/bit. Since the free-carrier accumulation in the MOS
 only depends on the D field in the gate insulator, the
 performance of the hybrid silicon-ITO modulator can be
 further improved with high- k materials such as HfO $_2$. For
 example, if we replace the 20 nm SiO $_2$ with 5 nm thick HfO $_2$,
 the applied voltage will be reduced to 1 V to achieve the same
 D field using current 12 V bias. In this case, the RC-limited
 speed will decrease to 40 GHz due to the increased capacitance.
 However, the resonance tuning efficiency will increase to 360
 pm/V, and the energy consumption will drop to 6.2 fJ/bit. In
 addition, our current hybrid silicon-ITO nanocavity modulator
 only possesses a moderate Q -factor of 1000 due to our
 fabrication quality such as the surface roughness and the
 deviation of the air hole diameters. Through advanced
 designs⁴² and optimized fabrication, a PC nanocavity with
 higher Q -factor is achievable. We anticipate that both the ER
 and the operation voltage will be improved in further
 development, offering the possibility to achieve hundreds of
 attojoule/bit energy efficiency in the future. For example, if the
 Q -factor is improved to 5000 (Q -factor-limited bandwidth will
 be 240 GHz), we can further reduce the operational voltage by
 5 \times and improve the energy efficiency by 25 \times to 250 aJ/bit.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the
 ACS Publications website at DOI: 10.1021/acs.nano-
 lett.7b04588.

Calculation of permittivity and refractive index of ITO
 and Si, details of electrical modeling of ITO/oxide/Si
 capacitor, optical simulation, calculation of the capaci-
 tance and resistance, experimental details of device
 fabrication, and measurement setup (PDF)

AUTHOR INFORMATION

Corresponding Author

*E-mail: wang@oregonstate.edu.

ORCID

Alan X. Wang: 0000-0002-0553-498X

Author Contributions

A.X.W. and R.T.C. conceived the ideas of the project. E.L.
 performed the simulations and devised the geometry of the
 modulators. E.L. and Q.G. fabricated the hybrid Si-ITO
 modulators. E.L. conducted the optical and electrical character-
 ization of the modulators under the supervision of A.X.W. All
 authors discussed the results. E.L. and A.X.W. cowrote the
 paper. R.T.C. and A.X.W. supervised the project.

Notes

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